



ELSEVIER

Journal of Nuclear Materials 258–263 (1998) 537–542

Journal of  
nuclear  
materials

# Study on interaction of hydrogen isotopes with radiolysis products in lithium oxide

Viktors Grišmanovs, Masaki Taniguchi, Satoru Tanaka \*, Toshiaki Yoneoka

*Department of Quantum Engineering and Systems Science, The University of Tokyo, 7-3-1 Hongo, Bunkyo-ku, Tokyo 113, Japan*

## Abstract

Out-of-pile tritium release experiments were performed for  $\text{Li}_2\text{O}$  polycrystalline powder (2–4 mm) irradiated by 2.2 MeV electrons to various absorbed doses up to 100 MGy and by thermal neutrons with a fluence of  $10^{17}$  neutron/m<sup>2</sup>. The linear temperature increase of the electron-irradiated samples showed two tritium release peaks: first started at around 600 K with a maximum at 800 K and second appeared at around 950 K with the maximum approximately at 1200 K. It is thought that the tritium release at high temperatures is due to the thermal decomposition of LiT. The formation of LiD under the electron irradiation was investigated by using the diffuse-reflectance Fourier transform infrared (FTIR) absorption spectroscopy. The  $\text{Li}_2\text{O}$  powder was irradiated by electron accelerator under  $\text{D}_2$  containing atmosphere ( $\text{N}_2 + 10\% \text{D}_2$ ). A specific absorption band for the  $\text{Li}_2\text{O}$  sample was observed at  $668 \text{ cm}^{-1}$  and attributed to the Li–D stretching vibration. © 1998 Elsevier Science B.V. All rights reserved.

## 1. Introduction

Lithium oxide has been suggested as a suitable breeding blanket material for a fusion reactor. In an operating fusion reactor, the tritium breeding blanket should reach a steady-state condition whereby the tritium release rate equals the production rate. The tritium release must be fast enough so that the tritium inventory in the blanket does not become excessive. Slow tritium release will result in a large tritium inventory, which is unacceptable from the both economic and safety viewpoints.

The irradiation damage in  $\text{Li}_2\text{O}$  single crystals exposed to a high fluence with fast neutrons ( $3.9 \cdot 10^{26}$  neutron/m<sup>2</sup>) at about 650 K has been investigated by methods of electron spin resonance and optical absorption in the BEATRIX-II program [1]. The electron spin resonance data have been interpreted to indicate the presence of F-center aggregates in  $\text{Li}_2\text{O}$  single crystals with 1.8% and 0.07%  $^6\text{Li}$  concentration. F-center aggregates are nothing more nor less than Li-atom aggregates which consist from the Li-atom ions

surrounded by the F-centers [2]. The integral charge of such a cluster is negative. In most of the blanket designs using the  $\text{Li}_2\text{O}$ , the  $^6\text{Li}$  enrichment of  $\text{Li}_2\text{O}$  is high (about 50–60%) and the neutron fluence at the lifetime is expected to be higher than that reached in the experiments performed by Noda et al. [1], and hence a high concentration of F-center aggregates could be created in the blanket. The operating temperatures for a typical blanket design using  $\text{Li}_2\text{O}$  are in the range of 720–920 K [3]. The F-center aggregates will remain in  $\text{Li}_2\text{O}$  in the lower temperature part of the blanket, since they are completely recovered at around 923 K [4]. This can cause a tritium inventory due to its trapping by Li-atom aggregates. Thus, the irradiation tests up to a high fluence at temperatures in the lower temperature part of the blanket are required.

The research on the influence of radiation defects on tritium release behavior from polycrystalline  $\text{Li}_2\text{O}$  [5] has revealed that the radiation defects in  $\text{Li}_2\text{O}$  cause the retention of tritium. It has been thought that the tritium recovery is affected by the formation of Li–T bond, which is tolerant of high temperatures; for instance, the thermal decomposition of Li–H bond is estimated at 1223 K [6]. The retardation of tritium decreased with increasing absorbed dose in the range from 50–140 MGy. In order to confirm the tritium retention observed

\* Corresponding author. Tel./fax: +81-3 5800 6860; e-mail: chitanak@q.t.u-tokyo.ac.jp.

in in-pile experiments [5], the out-of-pile tritium release measurements were performed in the present work. The experiments were carried out for the polycrystalline  $\text{Li}_2\text{O}$  powder heated up to 1273 K.

To evaluate the temperature efficient for the formation of Li-T bond the following simulation was carried out. The  $\text{Li}_2\text{O}$  powder was irradiated by an electron accelerator under  $\text{D}_2$  containing atmosphere ( $\text{N}_2 + 10\% \text{D}_2$ ) at various temperatures, and the formation of LiD was investigated by using the diffuse-reflectance FTIR absorption spectroscopy.

## 2. Experimental

The out-of-pile tritium release experiments were performed using a polycrystalline  $\text{Li}_2\text{O}$  of the particle size of 2–4 mm. The sample was 99.9% pure (Al: 0.02%, Ca: 0.05%, Cu: 0.001%, Fe: 0.01%, Si: 0.01%, Sn: 0.03%). The specimens were sealed in quartz containers filled with air atmosphere and irradiated by accelerated electrons (energy – 2.2 MeV, dose rate – 8 kGy/s, temperature – around 370 K) to absorbed dose up to 100 MGy at the DYNAMITRON facility of the University of Tokyo.

Afterwards, the specimens were exposed at about 310 K to the fluence of  $10^{17}$  thermal neutron/ $\text{m}^2$  at the research reactor “YAYOI” of the University of Tokyo. The samples without prior irradiation by electrons were exposed to the same fluence of thermal neutrons and used as reference samples. The out-of-pile tritium release experiments were carried out during the temperature transient from 293 to 1273 K. The temperature ramp

was 10 K/min. The released tritium was removed by the sweep gas of  $\text{N}_2$  and detected by the ionization chamber.

The formation of LiD under the electron irradiation was investigated by using diffuse-reflectance FTIR absorption spectroscopy. The  $\text{Li}_2\text{O}$  powder (48–65 mesh) was irradiated by electron accelerator under  $\text{D}_2$  containing atmosphere ( $\text{N}_2 + 10\% \text{D}_2$ ) at around 373, 673 and 893 K. For the reference sample we used the annealed  $\text{Li}_2\text{O}$  powder, which was examined by diffuse-reflectance FTIR absorption spectroscopy under Ar and Ar +  $\text{D}_2$  atmospheres.

## 3. Results and discussion

### 3.1. Out-of pile tritium release

Fig. 1 shows the tritium release curve of the polycrystalline  $\text{Li}_2\text{O}$  specimen exposed at around 310 K to the fluence of  $10^{17}$  thermal neutron/ $\text{m}^2$ . The sharp peak of released tritium can be seen at around 750 K. The experimental data indicate that there is no influence of radiation defects on tritium release. The absorbed dose for this specimen is estimated at around 10 kGy. Thus, the formation of simple F-centers proceeds at such low absorbed doses [5,7]. It seems likely that the interaction of F-centers with tritium at low irradiation dose is negligible.

The irradiation of  $\text{Li}_2\text{O}$  specimens by 2.2 MeV electrons to an absorbed dose of 75 MGy followed with the fluence of  $10^{17}$  thermal neutron/ $\text{m}^2$  causes the change of the tritium release kinetics (Fig. 2). One can recognize two broad peaks in the tritium release curve: the first

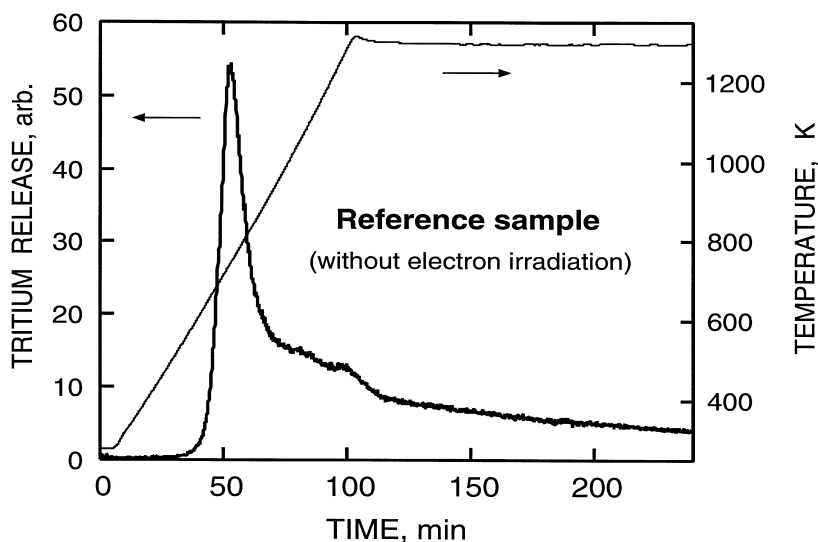


Fig. 1. Out-of-pile measurement of tritium release from the  $\text{Li}_2\text{O}$  specimen irradiated by thermal neutrons to the fluence of  $10^{17}$  neutron/ $\text{m}^2$ .

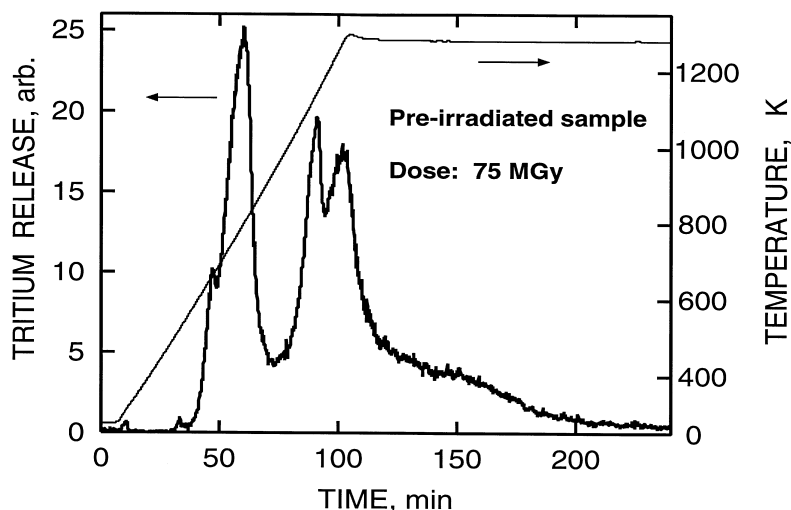


Fig. 2. Out-of-pile measurement of tritium release from the  $\text{Li}_2\text{O}$  specimen pre-irradiated by 2.2 MeV electrons to the absorbed dose of 75 MGy and by thermal neutrons to the fluence of  $10^{17}$  neutron/m<sup>2</sup>.

starts at 600 K with a maximum at 800 K and the second appears at 950 K with two maxima at 1150 and 1270 K. The first broad peak is associated with (1) desorption of physically and chemically adsorbed tritium on the sample surface, (2) decomposition of LiOT, and (3) migration of tritium dissolved in the bulk. It seems that the tritium bonded in the form of LiT is not released below 850 K as a significant fraction. The second peak begins to appear at around 950 K which is coincident with the phase transition of LiH (the 961.8 K corresponds to the crystal  $\rightarrow$  liquid transition [8]). The second broad peak is thought to be due to the tritium bonded in Li-T bond, which was formed by the reaction of colloidal Li with tritium. The presence of the two maxima at about 1150 and 1270 K in this peak can be explained from the following viewpoint. The highest concentrations of radiation defects and tritium are in layers adjacent to the surface [9]. Thus, the first peak at 1150 K can be associated with tritium released due to the thermal decomposition of LiT at the near surface layers of the sample. Consequently, the second peak at 1270 K is the tritium released from the bulk of the sample where it was locally trapped at colloidal centers.

Very similar results were obtained in our preliminary investigation on the influence of radiation defects on tritium release from  $\text{Li}_2\text{O}$  single crystals [10]. The tritium release curve of the sample pre-irradiated with 2.2 MeV electrons to an absorbed dose of 50 MGy and followed by exposure to thermal neutrons with a fluence of  $10^{17}$  n/m<sup>2</sup> consisted of a lot of small peaks, which were thought to be associated with tritium trapped by radiation defects at a different distance from the surface of the sample. One can suggest that the distribution of lithium particles created by irradiation is inhomogeneous. Be-

cause under irradiation the colloidal particles are formed by agglomeration of F-centers, the larger the colloids the larger is the cross section for the capture of new centers. This type of growth leads to a logarithmic-normal distribution of colloids size [11]. Such a distribution is determined by the grain size distribution or by the size distribution of cavities and dislocations created by irradiation [12]. This indicates an effective formation of colloidal particles in a certain region of matrix where cavities and dislocations are produced by irradiation. As a result, the tritium is locally trapped by lithium colloids in the form of LiT, which is thermally stable at high temperatures up to 950–1200 K. This effect is undesirable from the tritium inventory standpoint. However, the generation of tritium by thermal neutron irradiation was conducted at about 310 K, which is different from the operating temperature of the blanket of a fusion reactor.

In the paper by Vajda et al. [4], the recovery of metallic Li colloids and F<sup>+</sup> center agglomerates was reported to occur at around 673 and 923 K, respectively. F-center aggregates (in other words: clusters or agglomerates) are nothing more or less than Li-atom aggregates which consist of Li-atom ions surrounded by F-centers. Thus, in any case the formation of Li-T bond can occur in  $\text{Li}_2\text{O}$  even at high temperatures. The problem only is the amount of formed LiT. The concentration of F<sup>+</sup> center clusters determines the accumulation rate of LiT, and in its turn depends on temperature and irradiation dose. Thus, a certain region of temperature exists wherein the formation of Li-T bond is the most effective. In fact, the data of the ESR examination of  $\text{Li}_2\text{O}$  single crystals irradiated at 650 K by fast neutrons in the BEATRIX II [1] showed that the

$F^+$  center aggregates is created in  $Li_2O$ . According to our investigations [13], 650K can be a critical temperature for the formation of Li colloids. However, there is no clear understanding about the optimum temperature for the formation of  $F^+$  center aggregates. Thus, the investigation of the influence of the radiolysis products on tritium release kinetics from  $Li_2O$  under the conditions expected in the thermonuclear reactor is desirable.

### 3.2. The observation of Li–D stretching vibration

In order to simulate the formation of Li–T bond under irradiation conditions, the following approach was applied. The  $Li_2O$  powder was exposed to accelerated electrons under  $N_2 + 10\% D_2$  atmosphere at various temperatures. The absorbed dose was close to the range wherein the formation of colloidal Li can take place [13]. In that research, we found that the formation of colloidal Li in polycrystalline  $Li_2O$  under electron irradiation at 380 K occurred at an absorbed dose of about 100 MGy. It was presumed that the formed colloidal Li formed during electron irradiation reacted with  $D_2$  absorbed in  $Li_2O$  powder to form the Li–D bond.

The samples prepared in the way described above were examined by using diffuse reflectance method of FTIR absorption spectroscopy. This method was successfully applied to the research on  $Li_2O$  surface nature [14,15]. Two absorption peaks of surface –OH groups were observed at 3677 and 3565  $cm^{-1}$ , and three stretching vibration peaks of surface –OD groups were detected at 2748, 2717 and 2696  $cm^{-1}$  [14]. The stretching vibration at 1218  $cm^{-1}$  was attributed to Li–H bond [15].

Fig. 3 shows the absorption spectra of  $Li_2O$  irradiated by electrons to adsorbed dose of 100 MGy at about 373 K under  $N_2 + 10\% D_2$  atmosphere. The stretching vibration peak can be seen at around 668  $cm^{-1}$ . This peak is detected the first time for  $Li_2O$ , and it is attributable to the Li–D bond. The absorption peak at 668  $cm^{-1}$  has been assigned to the Li–D bond due to two reasons: firstly, this peak appears in the FTIR spectra only under the  $D_2$  containing atmosphere, and secondly, the shift to a lower wave number was observed for LiD in comparison with LiH. The latter observation is a common feature for the stretching vibration of chemical bonds with different hydrogen isotopes. For instance,

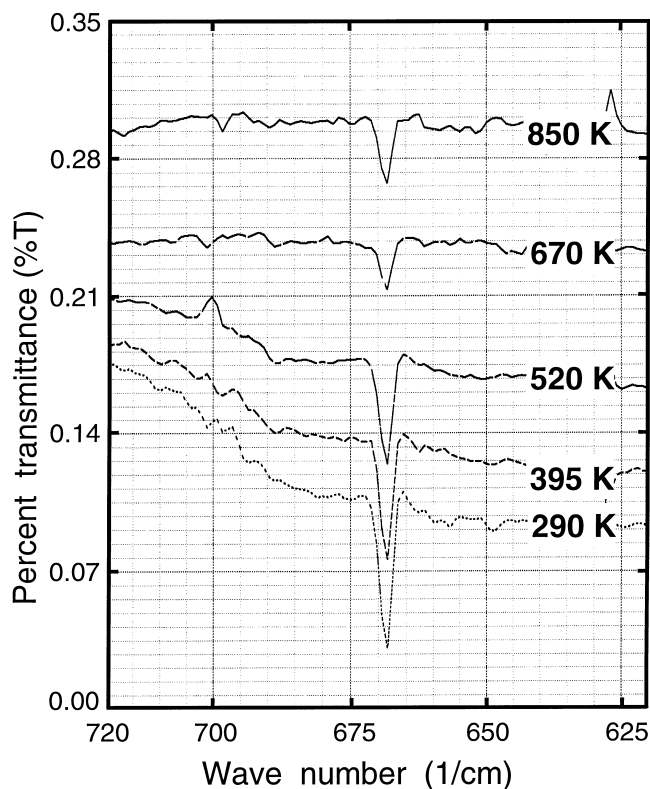


Fig. 3. Temperature dependence of absorption spectra of the  $Li_2O$  specimen irradiated at 373 K by 2.2 MeV electrons to the absorbed dose of 100 MGy under a  $N_2 + 10\% D_2$  atmosphere.

the absorption peaks for the pure LiH [15] and LiD [16] have been observed at 1280 cm and 880 cm<sup>-1</sup>, respectively.

The intensity of the 668 cm<sup>-1</sup> peak decreases with temperature up to 850 K (Fig. 3), but it reduces to almost the same value while the temperature was decreased. This fact indicates the thermal stability of the 668 cm<sup>-1</sup> absorption peak. The irradiation of Li<sub>2</sub>O powder by electrons at 673 and 893 K under N<sub>2</sub> + 10% D<sub>2</sub> atmosphere also resulted in the appearance of the stretching vibration peak at 668 cm<sup>-1</sup>. However, the intensities of absorption peaks are lower than that detected for the sample irradiated at 373 K. In such a way, this testifies that the formation of Li–D bond is less efficient at high temperatures. The observed data suggest that the formation of Li–D bond can occur up to 893 K. However, a possibility of the production of Li–D bond at a higher temperature than 893 K is a subject to be studied.

The research on the FTIR absorption spectra of annealed Li<sub>2</sub>O under Ar and Ar + D<sub>2</sub> atmospheres disclosed that the formation of Li–D bond occurs on the sample surface in the latter case. The absorption peak at 668 cm<sup>-1</sup> observed for annealed Li<sub>2</sub>O sample (Fig. 4) is

coincident with that detected for irradiated by electron specimens under N<sub>2</sub> + 10% D<sub>2</sub> atmosphere (Fig. 3). This fact does not allow us to conclude with confidence how the formation of Li–D bond proceeds in Li<sub>2</sub>O during electron irradiation under atmosphere containing D<sub>2</sub>: (1) due to the interaction with colloidal Li or (2) by reactions on the surface of Li<sub>2</sub>O. It seems that the formation of Li–D bond easily takes place just in the presence of D<sub>2</sub> in the purge gas during the measurement of the diffuse-reflectance FTIR absorption spectrum.

#### 4. Conclusions

It has been shown that the radiation defects introduced by electron irradiation cause the retention of tritium due to its bonding in the thermally stable state. The interaction of tritium with colloidal Li produced by electron irradiation is thought to result in the formation of Li–T bond. The tritium release at high temperatures (>950 K) indicates the decomposition of Li–T bond.

The formation of Li–D bond on the surface of Li<sub>2</sub>O under the atmosphere containing D<sub>2</sub> was observed by diffuse-reflectance FTIR absorption spectroscopy for the

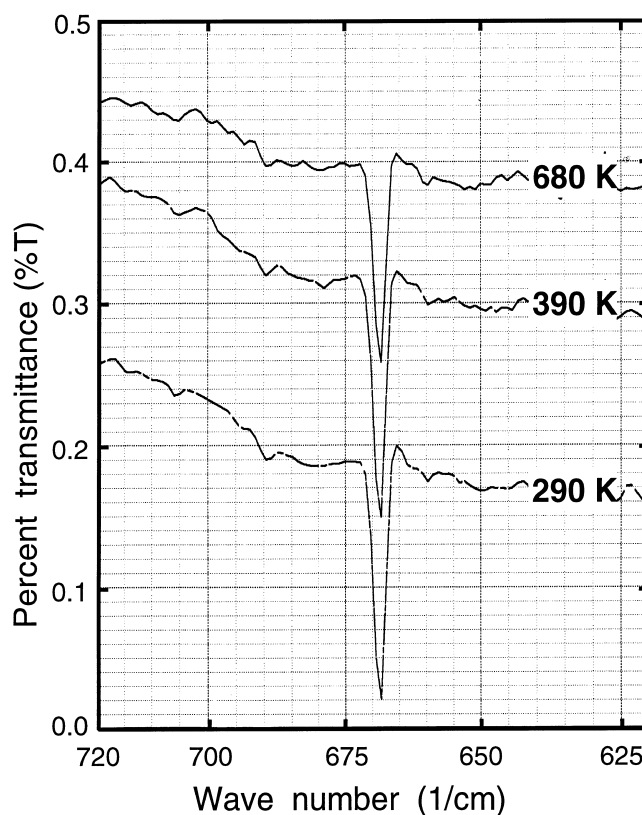


Fig. 4. Temperature dependence of absorption spectra of a non-irradiated Li<sub>2</sub>O specimen under an Ar+D<sub>2</sub> atmosphere.

first time. However, a mechanism of the production of Li–D bond under electron irradiation in the presence of D<sub>2</sub> in the purge gas is unclear: the binding of D<sub>2</sub> in Li–D can proceed by surface reactions as well as by interaction of D<sub>2</sub> with colloidal Li produced by irradiation.

## References

- [1] N.M. Masaki, K. Noda, H. Watanabe, R.G. Clemmer, G.W. Hollenberg, *J. Nucl. Mater.* 212 (1994) 908.
- [2] K. Schwartz, J. Ekmanis, *Dielectric Materials: Radiation Processes and Radiation Stability*, Zinatne, Riga, 1989 (in Russian).
- [3] H. Takatsu, S. Mori, H. Yoshida, T. Hashimoto, T. Kurasawa, K. Koizumi, M. Enoda, S. Satoh, T. Kuroda, T. Suzuki, K. Ioki, T. Kanazawa, in: *Proceedings of SOFT-17*, Roma, Italy, 1992, p. 1504.
- [4] P. Vajda, F. Beuneu, these Proceedings.
- [5] V. Grishmanov, S. Tanaka, J. Tiliks, G. Kizane, A. Supe, T. Yoneoka, *Fusion Engrg. Design* (in press).
- [6] I. Barin, *Thermodynamical Data for Pure Substances*, VCH, New York, 1989.
- [7] V. Grishmanov, S. Tanaka, J. Tiliks, in: *Proceedings of SOFT-19*, Lisbon, Portugal, 1996, p. 1451.
- [8] M.W. Chase et al., *J. Phys. Chem. Ref. Data* (Suppl. 1) (1985) 1224–1227.
- [9] A. Abramenkovs, J. Tiliks, G. Kizane, V. Grishmanov, A. Supe, *J. Nucl. Mater.* 248 (1997) 116.
- [10] V. Grishmanov, S. Tanaka, T. Yoneoka, *J. Nucl. Mater.* 248 (1997) 128.
- [11] A. Berthault, S. Bedere, J. Matricon, *J. Phys. Chem. Solids* 38 (1977) 913.
- [12] R.T. Dehoff, *Trans. Met. Soc. AIME* 230 (1965) 25.
- [13] V. Grishmanov, S. Tanaka, J. Tiliks, G. Kizane, A. Supe, L. Grigorjeva, *Nucl. Instr. and Meth. Phys. Res. B* 134 (1998) 27.
- [14] S. Tanaka, M. Taniguchi, M. Nakatani, D. Yamaki, M. Yamawaki, *J. Nucl. Mater.* 218 (1995) 335.
- [15] J.P. Kopazs, J. Ortiz-Villafuerte, C.E. Johnson, in: *Proceedings of the Fourth Symposium on the Fabrication and Properties of Fusion Ceramics*, 1995, p. 116.
- [16] M.H. Brodsky, E. Burstein, *J. Phys. Chem. Solids* 28 (1965) 1655.